

# Infrared Band-Shape Studies of Sulphate Glasses \*

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IR band-shape analysis has been carried out on the  $620\text{ cm}^{-1}$  deformation band of the sulphate ion in several  $\text{Na}_2\text{SO}_4\text{-K}_2\text{SO}_4\text{-ZnSO}_4$  glasses. Variations of correlation times and second moments suggest that reorientational motions of sulphate ions begin to evolve prior to the glass-transition temperature. The correlation times may support a cluster model for the glass-transition.

## Introduction

Several vibrational band-shape studies which have used either Raman or infra-red (IR) spectroscopy have been reported in the literature [1–3]. With very few exceptions [4, 5], however, the studies have generally been performed on symmetrical molecules in non-viscous fluids. In such situations molecular motions may be characterized with little ambiguity [6].

Translational and rotational motions of groups generally lead to a broadening of the vibrational bands. It is in fact possible to show that the band-shape ( $I(\omega)$ ) is the Fourier transform of the correlation function,  $\varphi(t)$ , of the transition dipole moment at times 0 and  $t$ .

$I(\omega)$  can be expressed as

$$I(\omega) = \int_{-\infty}^{+\infty} \exp(-i\omega t) \langle \mathbf{u}(0) \cdot \mathbf{u}(t) \rangle dt, \quad (1)$$

where  $\mathbf{u}(0)$  and  $\mathbf{u}(t)$  are the transition dipole moment vectors at times 0 and  $t$ , respectively.  $\langle \mathbf{u}(0) \cdot \mathbf{u}(t) \rangle$  represents the correlation function  $\varphi(t)$  and can be obtained [7] by Fourier inversion of (1):

$$\varphi(t) = \langle \mathbf{u}(0) \cdot \mathbf{u}(t) \rangle = \int_{\text{band}} \hat{I}(\omega) \exp(i\omega t) d\omega,$$

where

$$\hat{I}(\omega) = I(\omega) / \int_{\text{band}} I(\omega) d\omega.$$

$\varphi(t)$  is a measure of the average variation of the angle between  $\mathbf{u}(0)$  and  $\mathbf{u}(t)$  between times 0 and  $t$ ; such a variation could be caused by molecular reorientation. However, relaxation through vibra-

tional excited states can also affect  $\varphi(t)$ . Therefore the total dipole correlation function can be written [8, 9] as

$$\varphi(t) = C_{\varphi}^{\text{R}}(t) C_{\varphi}^{\text{V}}(t), \quad (2)$$

where  $C_{\varphi}^{\text{R}}(t)$  and  $C_{\varphi}^{\text{V}}(t)$  are the rotational and vibrational correlation functions, respectively.  $C_{\varphi}^{\text{R}}(t)$  in (2) includes the effects of both rotational and roto-diffusive motions. Separation of  $C_{\varphi}^{\text{R}}(t)$  and  $C_{\varphi}^{\text{V}}(t)$  is accomplished through Raman spectroscopy [3]. IR band-shape analyses cannot unambiguously separate  $C_{\varphi}^{\text{R}}(t)$  and  $C_{\varphi}^{\text{V}}(t)$ .

In a situation like the glass transition, which corresponds to the onset of diffusive motion manifested by viscous flow, we should expect that band-broadening arises from rotational and roto-diffusional motion rather than from vibrational relaxation. Therefore, we thought that it would be instructive to analyse IR band-shapes assuming that vibrational band-broadening is not significant as compared to the effect of particle reorientation.

In this laboratory we have investigated sulphate glasses in considerable detail [10, 11]. These glasses are ionic and contain discrete  $\text{SO}_4^{2-}$  ions. Therefore the vibrational bands of these ions may be used for such investigation in the light of the foregoing remarks.

## Experimental

Preparation of these glasses has been described in detail elsewhere [10]. For IR studies, a sample — KBr mixture was ball-milled for  $\sim 15$  mins and pelletised under 7 kb pressure. The resultant pellets were transparent at ambient temperature but tended to opacify at higher temperatures. IR spectra were recorded on a Perkin-Elmer 580 spectrometer using a SPECAC variable-temperature

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cell and temperature controller. Temperature stability was good to  $\pm 1$  K.

The sulphate deformation mode ( $\nu_4$ ) [12] at  $620\text{ cm}^{-1}$  was chosen for band-shape analysis as it was relatively free of side-bands. Care was taken to record intensities in the wings of the band. The baseline was drawn by visual inspection. Errors using this approach did not seem to be serious, as repeated transforms using different likely baselines yielded correlation functions little different from one another. Fourier transformation was carried out on a DEC-1090 system using only that half of the band which was free of side-band. Intensities were measured at  $2\text{ cm}^{-1}$  intervals in order to perform numerical integration.

The full-width at half-maximum (FWHM) was measured both for the complete band and also by doubling the width of the side-band-free half of the band. There was no significant difference between the two values so obtained since the side-band reaches maximum intensity only in the wing of the  $\nu_4$  band. The FWHM values that we have measured agree well with Raman and IR results on other crystalline and molten sulphate systems [13, 14]. Second moments,  $M_2$ , were calculated using the formula

$$M_2 = \int_{\text{band}} (\omega - \omega_c)^2 I(\omega) d\omega.$$

## Results and Discussion

The  $620\text{ cm}^{-1}$  band used is shown for a typical glass in Figure 1. Figure 2 shows typical  $\varphi(t)$  vs  $t$  plots of a single glass for a few temperatures. Curvature of  $\varphi(t)$  at small  $t$ , indicative of relatively free

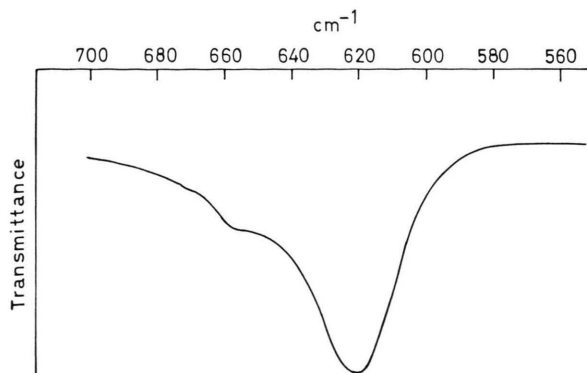


Fig. 1. A typical  $620\text{ cm}^{-1}$  band that was used in the Fourier transformation.

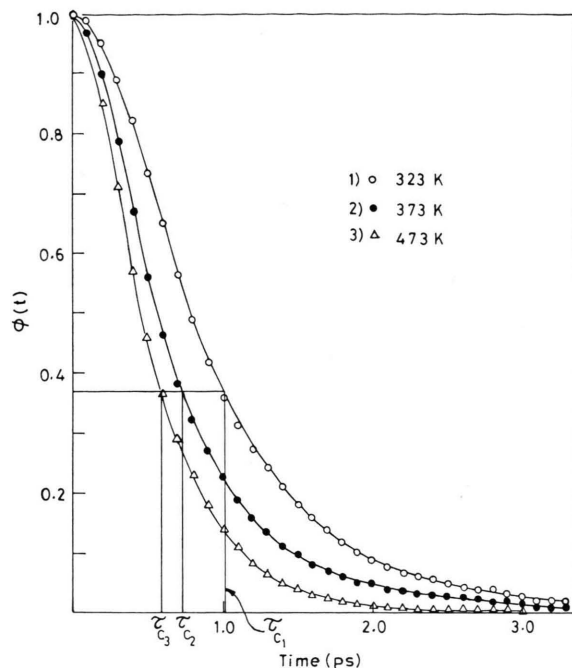


Fig. 2. Correlation function  $\varphi(t)$  as a function of time ( $t$ ) for the  $50\text{ZnSO}_4\cdot 40\text{K}_2\text{SO}_4\cdot 10\text{K}_2\text{SO}_4$  glass for three temperatures.  $\tau_c$  is indicated for each temperature.

rotation at short times, is clearly seen in Figure 2.  $\tau_c$  values are also shown on the  $\varphi(t)$  plots. Figure 3 shows the variation of these  $\tau_c$  values as a function of  $T/T_g$  (where  $T_g$  is the calorimetric glass transi-

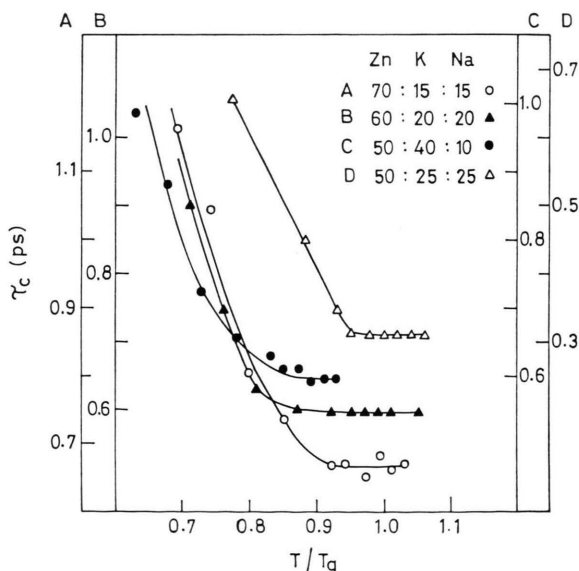


Fig. 3. Correlation time  $\tau_c$  as a function of  $T/T_g$  for glasses of the indicated compositions.

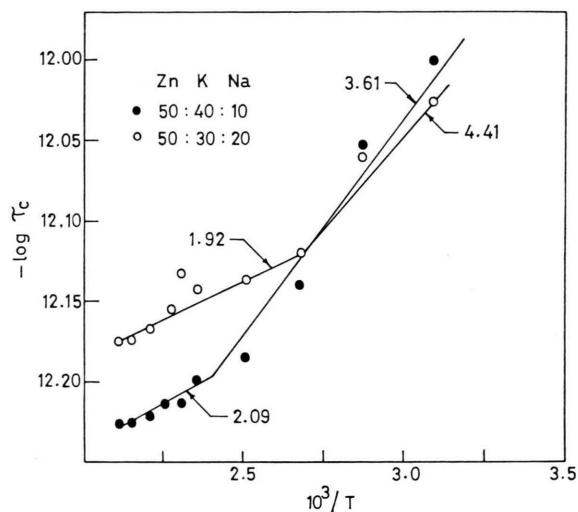


Fig. 4.  $\log \tau_c$  against  $10^3/T$  for two glass compositions. The figures indicate activation energies in  $\text{kJ mol}^{-1}$ .

tion temperature) for various glass compositions.  $\tau_c$  decreases smoothly and levels off around  $T/T_g \sim 0.9$ .  $\tau_c$  was fitted to an Arrhenius-type equation. Typical  $\log \tau_c$  against  $10^3/T$  plots are shown in Figure 4. A sharp bend is observable in the glassy region ( $T/T_g < 1$ ). The activation energy ( $E_a$ ) in this region is high. Above the glass transition ( $T/T_g > 1$ ) a lower activation energy is observed. The  $E_a$ 's in both regions are indicated in Figure 4.  $E_a$  is of the order of 5–10  $\text{kJ mol}^{-1}$  in the glass, which is quite reasonable [7], and decreases by 40% near  $T_g$ . Second moments in a few glasses were evaluated and plotted as a function of  $T/T_g$  in Figure 5. There is an initial rise after which the second moments level off at  $T/T_g \sim 0.9$ .

The decrease in  $\tau_c$  and the increase in second moment as a function of temperature may be considered as indications of the onset of reorientational motion around  $T_g$ . Since we have assumed that vibrational relaxation is relatively unaffected by temperature, we prefer to attribute the decrease in  $\tau_c$  to reorientational motions.

We suggest that the rapid decrease in  $\tau_c$  (uncorrelation) may arise from hindered rotation coupled with translational motion. It is possible that the anions acquire a state of large amplitude libration or possibly free rotation (roto-diffusion) as suggested by the behaviour of second moments in Figure 5. For  $T/T_g > 1$  it is likely that saturation values of rotation energies are reached. We cannot

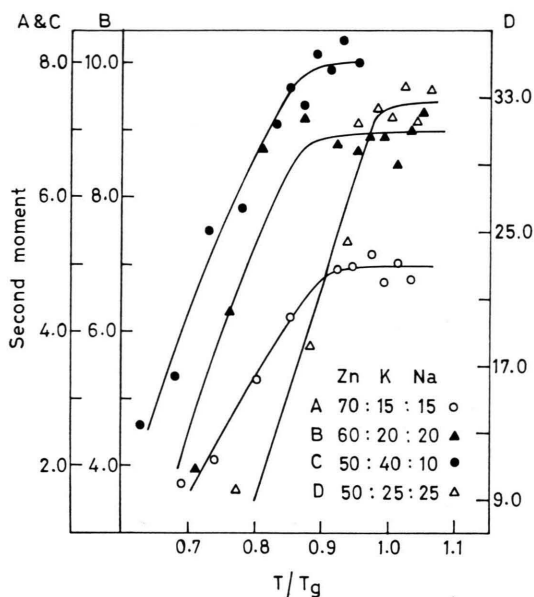


Fig. 5. The second moment as a function of  $T/T_g$  for four glass compositions: The Y-scales are for the indicated compositions.

however rule out the likelihood of motional narrowing of the band caused by such diffusional motion, but this is possibly more than compensated for by heterogeneous band broadening [5], leaving  $\tau_c$  effectively constant for  $T/T_g \geq 1$  (see Figure 3).

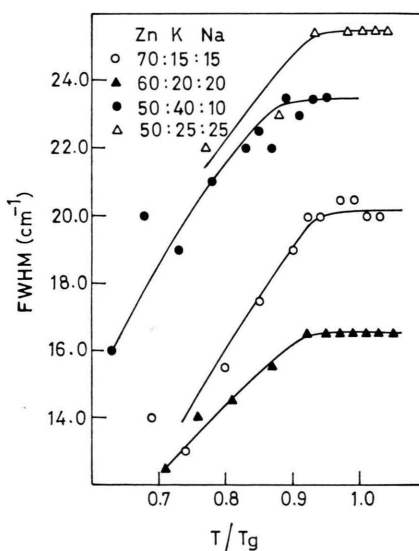


Fig. 6. FWHM of the  $620 \text{ cm}^{-1}$  band as a function of  $T/T_g$  for glasses of the indicated compositions.

This would accord well with the behaviour of FWHM itself as a function of temperature (Figure 6). FWHM increases as temperature increases and levels off around  $T/T_g \sim 0.9$ . The behaviour is similar to that of the second moments. It may be noted that an approximate correlation time  $\tau_c'$  is often defined as  $\tau_c' = 1/(2\pi x \text{FWHM})$ . We find that  $\tau_c'$  so determined is generally almost equal to  $\tau_c$ . We have thus tentatively attributed the levelling-off of  $\tau_c$  ( $T \geq T_g$ ) in Fig. 3 to two competing factors which become important at  $T/T_g > 1$ .

It is evident from Fig. 3 that  $\tau_c$  decreases by  $\sim 50\%$  around  $T/T_g \cong 1$ . It is interesting to note that  $\tau_c$  for the symmetrical  $\text{SO}_4^{2-}$  ion studied here is of the order of  $1 \times 10^{-12}$  s as compared to about  $2.5 \times 10^{-12}$  s for the unsymmetrical  $\text{NO}_3^-$  ion [5].

It is natural to expect that the process of re-orientational motion is sensitively affected by the potential field of the cations present at the  $\text{SO}_4^{2-}$  site. Figure 7(a) shows variations of  $\tau_c$  and  $E_a$  as a function of  $\text{ZnSO}_4$  concentration. The observed increase in  $\tau_c$  and decrease in  $E_a$  with increasing  $\text{ZnSO}_4$  content is to be expected on account of the local coulombic fields. However, the density also increases as the  $\text{ZnSO}_4$  content in glass increases. This may also contribute to the observed increase in  $\tau_c$ . In Fig. 7(b)  $\tau_c$  and  $E_a$  are plotted as a functions of  $\text{K}_2\text{SO}_4$  content in  $50\text{ZnSO}_4 \cdot x\text{K}_2\text{SO}_4 \cdot (50-x)\text{Na}_2\text{SO}_4$  glasses. An increase of  $\tau_c$  may

again be expected since  $\text{K}^+$  ions are likely to produce steric hindrance to  $\text{SO}_4^{2-}$  motion.

There is increasing evidence for the presence of clusters in glasses. In an earlier work using ESR spectra of an organic spin-probe in organic glasses [15–17], we observed that clustering begins to occur on the high-temperature side of  $T_g$  and is reflected in a sharp increase of spin-spin correlation times. In the present investigation the observed decrease in  $\tau_c$  as  $T_g$  is approached from the liquid could possibly originate from (a) melting of inter-cluster  $\text{SO}_4^{2-}$  ions or (b) from surfacial melting of clusters which causes loosening of  $\text{SO}_4^{2-}$  ions. In either event the present findings attribute the origin of  $\tau_c$  variations to intercluster material. Thus the observations are consistent with the cluster model for the glass-transition. Further, the “knee” temperature of Fig. 3 and the cluster formation “knee” temperature [18] in our earlier study help to understand, at least tentatively, the origin of the upper and lower temperature bounds of the glass transition region.

IR band shape studies of sulphate glasses therefore yield valuable information with respect to ionic motion. Though IR studies alone cannot quantitatively characterize all features of band broadening, temperature and chemical composition effects have been used in the present study in order to extract meaningful information on such particle motion particularly near  $T_g$ .

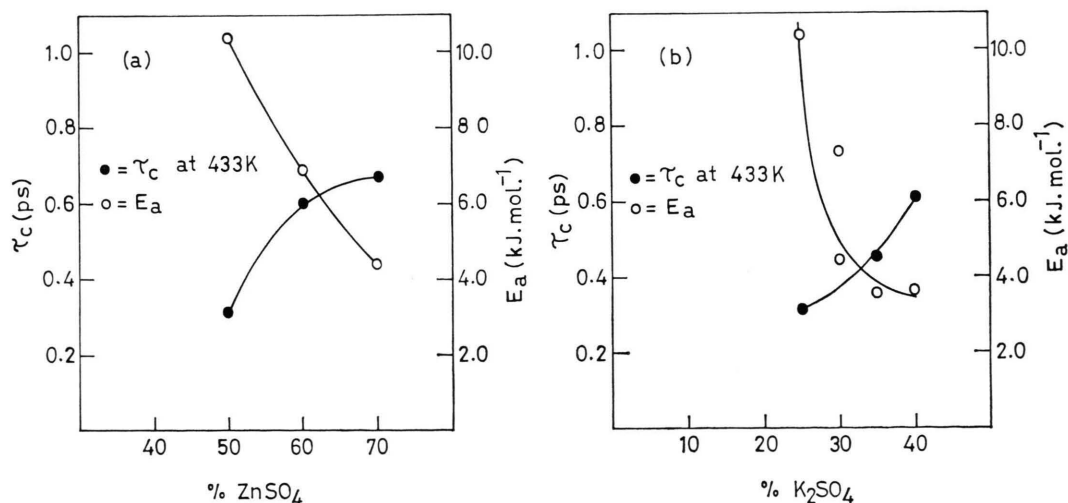


Fig. 7. (a)  $\tau_c$  and  $E_a$  as a function of  $\text{ZnSO}_4$  concentration, the remaining alkali being  $\text{Na}_2\text{SO}_4$ :  $\text{K}_2\text{SO}_4$ : 1:1. — (b)  $\tau_c$  and  $E_a$  as a function of  $\text{K}_2\text{SO}_4$  concentration for  $50\text{ZnSO}_4 \cdot x\text{K}_2\text{SO}_4 \cdot (50-x)\text{Na}_2\text{SO}_4$  glasses.

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